Microwave-assisted synthesis of graphene nanocomposites: recent developments on lithium-ion batteries

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Abstract: Lithium ion battery (LIB) is a popular power source for various portable mobile devices and even electrical vehicles. Graphene-based composites are important electrodes for LIBs due to their high-capacity, long cycle life, and impressive high-rate capability. Microwave-assisted synthesis is a promising approach to prepare graphene-based composites owing to its fast, energy-efficient features. By varying microwave irradiation conditions, surface functionality and morphology control can be tuned for either graphene or the introduced secondary phase in the graphene-based composites. When used for LIBs, the graphene-based composites can offer a variety of merits for the improved electrochemical properties such as facilitated lithium diffusion/storage and the increased mechanical stability of the electrodes during repetitive cycling. This article reviews the recent progress of microwave-assisted synthesis of graphene-based electrodes and their applications for LIBs. Graphene-supported transitional metal oxides anodes (Li-storage conversion mechanism), tin/germanium/silicon based anodes (lithium alloy mechanism), metal sulfides (conversion or lithium alloy mechanism), lithium-titanium-oxide-based anodes (lithium insertion mechanism), and graphene-decorated lithium iron phosphate cathodes are reviewed with more emphasis because these materials have attracted significant research concerns. The effect of microwave irradiation and the resultant structure and size control of graphene-based composites on their electrochemical properties is also elucidated.

Keywords: electrode, graphene, lithium ion batteries, microwave irradiation, nanocomposites

Introduction

With the development of electrical energy-storage materials to meet the increasing demand for the ever-growing energy consumption, lithium-ion battery (LIB), with excellence in terms of high energy density, no memory effect, long cycle life, and environmental friendliness, has been an attractive power source for portable mobile devices and stationary energy storage.1–4 As the commercial anode for LIBs, graphite gradually cannot satisfy the requirements of the ever-growing advanced high-power LIBs and new anode materials with higher energy density and power density must be explored.5–8 Since the discovery in 2004, graphene has attracted significant research concerns for various energy-relative applications including LIBs,9–11 solar cells,12–14 supercapacitors,15–17 and fuel cells.18–21 This is largely ascribed to its intriguing properties associated with the unique single-atom layered structure such as a large theoretical specific surface area of 2,600 m$^2$ g$^{-1}$,12 highly flexible but robust mechanical structure, and fast electronic conduction. As an anode for LIBs, graphene was reported with a reversible capacity of ~400–1,100 mAh g$^{-1}$,23–26 and its unsatisfactory cyclability has been mainly ascribed to its heavy agglomeration during cycling,
which leads to the loss of promising properties relative to the atomic-thickness structure. An effective strategy is to introduce the secondary phase to the graphene, which acts as a spacer to separate few-layer graphene nanosheets (GNS) and prevent their restacking to graphite platelets.\textsuperscript{27–29} The introduced component is usually also highly active for lithium-ion storage, and therefore, there is no capacity loss for the resultant composites. Moreover, the presence of graphene is also very beneficial for the improvement of the electrical conductivity and mechanical stability to the introduced secondary phase. As a result, graphene-supported composites can exhibit synergetic effect with respect to lithium-ion storage properties and achieve better electrochemical performance, especially long cycle life and impressive high-rate capability.\textsuperscript{30}

GNS used for fabrication of lithium ion anode materials were mostly synthesized through the chemical oxidation and reduction approach. The exfoliation and reduction of graphene oxide (GO) is the most important step to obtain GNS with suitable functionalities and reduction extents, which strongly affect the electrochemical properties of graphene or graphene-based composite electrodes. However, the exfoliation and reduction of GO rely heavily on the usage of strong reducing agents or a very high temperature. Most of these approaches are complicated, energy, and cost intensive. Recently, microwave exfoliation has been proved to be an attractive method for graphene preparation\textsuperscript{11–33} because it is a facile, time-efficient, and cost-effective process. Besides, the obtained graphene via microwave-assisted methods can exhibit larger average size, higher quality with residual functional groups, and better electrochemical properties for LIBs compared with those prepared from conventional methods.\textsuperscript{34–36}

Microwaves have been widely used in industrial applications such as food processing\textsuperscript{37,38} and industrial materials.\textsuperscript{39,40} Motivated by its advantages of facile, fast, secure, controllable, and energy-saving characteristics, microwave-assisted technique has achieved rapid development in the field of materials science.\textsuperscript{41,42} Microwave-assisted techniques such as solid-state microwave irradiation, microwave-assisted solvothermal/hydrothermal process can provide simple and fast routes to synthesize nanomaterials without high temperature or high pressure. Furthermore, the microwave technique is particularly useful for a large-scale synthesis without complicated preparation conditions.\textsuperscript{43–46} The rapid transfer of energy and fast decomposition of the precursors provided by microwave source would result in highly effective local reaction temperatures and significant enhancement in reaction rates. Besides, the microwave technique can provide an effective way to control particle size distribution and macroscopic morphology during the synthesis process because it can heat a substance uniformly and therefore a more homogeneous nucleation environment and a shorter crystallization time can be achieved compared to conventional heating.

In this review, we aim to investigate the mechanism of microwave-assisted syntheses of graphene and graphene-based nanocomposites, and summarize the recent development of graphene-supported nanocomposites for applications as electrodes for LIBs. Various types of graphene-based nanomaterials: mainly graphene-based transitional metal oxide anodes, tin/germanium/silicon based anodes, metal sulfides anodes, lithium-titanium-oxide-based anodes and graphene-decorated lithium iron phosphate cathodes are presented and discussed with respect to their morphological and size control in the microwave-assisted preparation process and their relation to the resultant lithium storage properties.

**Mechanism of microwave-assisted syntheses**

Microwave irradiation is an electromagnetic irradiation in the range of wavelengths from 0.01 m to 1 m with corresponding frequency range from 300 MHZ to 300 GHz.\textsuperscript{47} The domestic microwave generally owns a frequency of 2.45 GHz (a wavelength of 12.25 cm), while the industrial microwave usually owns two frequencies of 915 MHz and 2.45 GHz.\textsuperscript{47} Microwave has been widely used for heating those materials, which can absorb microwave energy and convert it into heat. In the presence of moisture or water, dielectric heating happens due to the dipolar nature of water. These permanently polarized dipolar molecules could rearrange in the direction of the electric field at a high speed, which would cause internal friction of molecules and further result in the volumetric heating of the whole material. Besides the dipolar mechanism, microwave heating may also occur due to the ionic mechanism, and oscillatory migration of ions in the material would generate heat under a high-frequency oscillating electric field.\textsuperscript{48} Consequently, microwave-assisted technology can provide a fast and effective approach to heat the material/system homogeneously from the interior. In contrast, traditional heating system, in which heat is transferred from the surface toward the center of the material under the help of heating mantle, water/oil bath or other external heat source, is relatively slow and inefficient.

GNS are usually obtained from graphite or GO, which is prepared by a modified Hummer’s method.\textsuperscript{49–51} The reduction of GO is usually carried out by chemical methods in the
presence of various hazardous reduction agents such as hydrazine and NaBH₄. In comparison, thermal treatment is a green method because no hazardous reduction agents are used. Instead of the conventional preparation of graphene in traditional heating system (furnace or oil bath), the eco-friendly microwave-assisted method has attracted increasing attentions in which the microwave-assisted solvothermal/hydrothermal methods can be adopted to treat GO or natural graphite in a microwave oven or microwave plasma-enhanced chemical vapor deposition (MPCVD) system. It is worth noting that the microwave exfoliation is an attractive and effective method for graphene synthesis from GO, in which GO is exfoliated with nontoxic solvents within a short reaction time of 1–15 minutes at a relatively low temperature range of 180°C–300°C. It was reported that the stable graphene suspension could be obtained from the GO suspension in an alkaline medium (pH = 10) or polar solvents (N,N-dimethylformamide, ethanol, 1-butanol, and water) in a facile microwave-assisted solvothermal process. Besides, the water-soluble polymer-grafted graphene sheets were prepared from GO in a household microwave oven at a power of 450 W for 4 minutes. The synthesis of three-dimensional (3D) nanostructure of “graphene nano-cup” anchored on the few layered graphene substrate under the microwave irradiation in a domestic microwave oven was reported by two steps: one-pot synthesis of graphene-coated metal nanoparticles anchored on the graphene sheets and the subsequent etching of metals. Furthermore, giant graphene sheets could be obtained by double microwave-assisted exfoliation of expandable graphite and highly hydrogenated graphene could be produced from GO by a one-step microwave irradiation process in hydrogen plasma, in which the deoxidation and concurrent hydrogenation were both achieved. A possible mechanism of graphene preparation by a microwave-assisted technology is illustrated in Figure 1. The microwave irradiation provides high local temperature and pressure atmosphere, and energy is transferred directly into the GO interior. Heat is produced from the interaction of irradiation with the polar bond of oxygen-containing functional groups on the surface and edge of GO sheets. Besides, the interaction between polar solvent and surface oxides on GO sheets is an important factor to determine the uniformity of deposits. Furthermore, the functional groups on the surface of GO are effectively reduced, and the reduction degree of graphene sheets is further improved.

The microwave-assisted technology for the synthesis of graphene has several obvious advantages. First, the microwave-assisted process is time-efficient without complicated synthesis procedure. Second, such process is cost-effective as the quantity of the used chemicals is greatly reduced compared to conventional approaches. Third, the average size of the obtained graphene from microwave-assisted technology can be ten times larger than those prepared by the conventional heating method. Finally, the graphene products from microwave-assisted technology are of high quality with controlled structure and residual functional groups.

Until now, microwave irradiation has been suggested as an effective tool to obtain carbon-relative composites with uniform dispersion and size and morphology control, because the microwave energy allows rapid heating and extremely rapid rate of crystallization to produce the desired nanocrystalline products. Meanwhile, during the microwave-assisted synthesis process, it is possible to control the growth of the favorable crystallographic plane by varying reaction time and the relative concentrations of different organic surfactants. Furthermore, the obtained nanostructures would extend from small spherical nuclei to short nanorod or nanosheet. As a result, various graphene-based nanocomposites with controlled size and shape, such as particle/crystal-on-sheet, nanorod/nanofiber-on-sheet, and nanosheet-on-sheet, can be obtained with the help of ecofriendly microwave-assisted technology, as illuminated in Figure 1. It is worthy noting that the hydrothermal/solvothermal processes operated in a single-mode microwave reaction or a multimode household microwave oven are the most used technology for the microwave-assisted synthesis of graphene-based nanocomposites. Nonuniform microwave is offered from multimode domestic microwave oven, in which

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**Figure 1** Schematic illustration of the synthesis of graphene and graphene-based composites with the assistance of microwave irradiation.
there is considerable variation in the microwave intensity throughout the reactor chamber. Moreover, the multimode domestic microwave oven can only roughly provide the time and several stages of power control (such as high, medium, low). In comparison, very uniform microwave can be generated in the specialized single-mode microwave reactor with stable microwave intensity in the chamber in which power, temperature, and time can be fine-tuned with continuous magnetic stirring. Therefore, the reaction environment is more uniform in the single-mode microwave reactor and better size and shape control should be achieved.

**GNS-supported transitional metal oxide anodes**

The microwave irradiation method has been applied for the synthesis of GNS-supported transitional metal oxide electrodes such as Co$_4$O$_6$-GNS, CuO-GNS, and Fe$_3$O$_4$-GNS. All these transitional metal oxides have approximately two to three times larger theoretical capacities than commercial graphite anode based on a well-known conversion mechanism of lithium storage. Lithium can reduce metal oxides to metal and form lithium oxide, and this reaction is reversible. The morphologies and lithium-storage properties of various graphene-supported transitional metal oxide anodes are summarized in Table 1. Among these GNS-supported transitional metal oxide anodes, Co$_4$O$_6$-GNS tends to form a particle-on-nanosheet morphology under microwave irradiation. As reported by Wang et al., two-dimensional (2D) porous Co$_4$O$_6$ nanosheets were obtained by a microwave solvothermal process at $180^\circ$C for 5 minutes (pressure: $7.5$ bar) in a single-mode microwave reactor (Nova, EU Microwave Chemistry, Shanghai, People’s Republic of China). As shown in Figure 2, these porous Co$_4$O$_6$ nanosheets have pore sizes of $60–100$ nm and a thickness around $100$ nm. After stacking with graphene, the Co$_4$O$_6$-GNS composite can form a sheet-on-sheet composite structure. The sheet-on-sheet composite shows superior Li-ion storage performances. Initial reversible charge capacity of $1,235$ mAh g$^{-1}$ is delivered, which decreases to $1,065$ mAh g$^{-1}$ after 30 cycles. This capacity is larger than those of GO or GNS, and even the theoretical value of pristine Co$_4$O$_6$ (890 mAh g$^{-1}$). The composite also exhibits an impressive good high-rate capability (a reversible capacity of $931$ mAh g$^{-1}$ at a large current rate of $5$ C [4,450 mA g$^{-1}$; $5$ C represents the current at which the cell capacity is charged/discharged in 1/5 h]).

Synthesized by the microwave-assisted technology, GNS-supported copper oxides can exhibit a variety of morphologies including zero-dimensional (0D) nanoparticle, one-dimensional (1D) nanowire/fusiform, and 2D nanoleaf/nanosheet morphologies. Among them, GNS-supported copper oxides with higher dimensional (1D and 2D) or core-shell morphology exhibit better electrochemical performance. By a fast single-mode microwave hydrothermal method, the GNS-supported sheet-like or fusiform-like CuO morphologies were obtained by varying the reaction temperature of microwave heating. As shown in Figure 3A–C, CuO-GNS sheet-on-sheet product was prepared at $170^\circ$C with CuO nanosheet of $0.3–0.5$ µm in size, while the CuO-GNS fusiform-on-sheet material was obtained at a lower temperature of $110^\circ$C with fusiform CuO product owning the length around $0.4–0.8$ µm and narrow tips. Figure 3D shows the electrochemical performance of the above two CuO-GNS composites in comparison with the physical mixture of CuO and GNS. Reversible charge capacities of 801 and 666 mAh g$^{-1}$ can be retained after 40 cycles for graphene-supported CuO nanosheet and fusiform composites, respectively. These reversible capacities are substantially larger than that of CuO-GNS (431 mAh g$^{-1}$) by a physical mixture after the same cycle numbers. As shown in Figure 3E, the graphene-supported CuO nanosheet composite exhibits an excellent rate capability with initial charge capacities of 981, 925, and 846 mAh g$^{-1}$ at 1, 2, and 5 C, respectively (1 C =700 mA g$^{-1}$).

The Fe$_3$O$_4$-GNS composite with porous nanorod-on-sheet morphology was synthesized by a solvothermal process in a microwave oven. The composite delivers a high reversible initial capacity of $1,016$ mAh g$^{-1}$ at $0.1$ A g$^{-1}$ (508 mAh g$^{-1}$ at 2 A g$^{-1}$ after 200 cycles). By comparison, the GNS-supported Fe$_3$O$_4$ nanoparticle composite obtained from the similar hydrothermal process in microwave oven exhibits better electrochemical properties with discharge capacities of 1,693, 1,142, 1,120, 1,098, and 1,027 mAh g$^{-1}$ in the first, tenth, 20th, 30th, and 50th cycles, respectively. A high capacity of $800$ mAh g$^{-1}$ is also observed at a current density of $800$ mA g$^{-1}$. Among the various GNS-supported Fe$_3$O$_4$ composites, Fe$_3$O$_4$ has a trend to form a particle or a porous particle morphology by the microwave-assisted technology. As reported by Yu et al, a series of Fe$_3$O$_4$-GNS nanostructures were synthesized by a simple nonaqueous solid-gel approach in a single-mode microwave apparatus. With different compositional ratios of Fe$_3$O$_4$ and GNS, reaction temperature, and times, as well as the synthesis method (Hummer’s method or modified Hummer’s procedure) of GO reactant, the obtained Fe$_3$O$_4$-GNS composites exhibit different particle sizes on the surface of GNS. The cycling performances of GNS-supported Fe$_3$O$_4$ composites are very stable even at high current...
density, which delivers high capacity of over 500 mA h g⁻¹ at 1,600 mA g⁻¹. Moreover, the Fe₂O₃–GNS rice-on-sheet and particle-on-sheet nanocomposites were synthesized by single-mode microwave hydrothermal technique with and without NH₄H₂PO₄, respectively, as shown in Figure 4A. The Fe₂O₃ nanorice is observed with a length of 200 nm and diameters in the range of ~40 nm in the middle part to only 3~5 nm in the tip (Figure 4C), while the nanoparticle is nearly a nanocube-like morphology of ~50~80 nm in size (Figure 4B). The Fe₂O₃–GNS rice-on-sheet composite exhibits large reversible charge capacities of 825, 762, and 633 mA h g⁻¹ at large currents of 1, 2, and 5 C (1 C = 1,000 mA g⁻¹) respectively. A high capacity of 582 mA h g⁻¹ can be observed at 1 C after 100 cycles. The rice-on-sheet composite also shows more stable cycle life and better high-rate performance than the particle-on-sheet composite (Figure 4D and E).

There are also several reports on GNS-supported other transitional metal oxides. Their electrochemical performances are summarized in Table 1. GNS-supported Mn₃O₄ particle composite was prepared by a microwave hydrothermal method. It exhibits a high specific capacity of more than 900 mA h g⁻¹ at 40 mA g⁻¹ and no capacity fading up to 50 cycles. By a similar microwave hydrothermal process, the MoO₃ nanobelt/graphene film was also reported by

Table 1 Summary on the morphologies and electrochemical performances of graphene-supported transitional metal oxide anodes

<table>
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<th>Morphologies</th>
<th>Electrochemical performances</th>
<th>Reference</th>
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<tr>
<td>3D Co₃O₄/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large discharge capacity of ~900 mA h g⁻¹ at 0.2 C, and the retained capacity of ~600 mA h g⁻¹ after 50 cycles</td>
<td>73</td>
</tr>
<tr>
<td>Co₃O₄/N-doped graphene</td>
<td>Nanoparticle-on-sheet</td>
<td>A reversible capacity of ~910 mA h g⁻¹ after 100 cycles at 100 mA g⁻¹</td>
<td>74</td>
</tr>
<tr>
<td>Co₃O₄/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A reversible capacity of ~1,785 mA h g⁻¹ after 90 cycles at 89 mA g⁻¹</td>
<td>75</td>
</tr>
<tr>
<td>Co₃O₄/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large charge capacity of ~934 mA h g⁻¹ at 70 mA g⁻¹ and a high retained capacity of 650 mA h g⁻¹ after 50 cycles at 700 mA g⁻¹</td>
<td>76</td>
</tr>
<tr>
<td>Co₃O₄/GNS</td>
<td>Sheet-on-sheet</td>
<td>An initial large charge capacity of ~1,235 mA h g⁻¹ at 89 mA g⁻¹, and the retained capacity of ~1,065 mA h g⁻¹ after 30 cycles</td>
<td>77</td>
</tr>
<tr>
<td>CuO/RGO</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large discharge capacity of ~1,043 mA h g⁻¹ at 0.1 mA cm⁻², and the retained capacity of ~516 mA h g⁻¹ after 45 cycles</td>
<td>78</td>
</tr>
<tr>
<td>CuO@Cu/RGO</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large charge capacity of ~734 mA h g⁻¹ at 50 mA g⁻¹, and the retained capacity of ~842 mA h g⁻¹ after 50 cycles</td>
<td>79</td>
</tr>
<tr>
<td>CuO–CuO₃O₄/GNS</td>
<td>Nanosphere-on-sheet</td>
<td>A reversible capacity of 487 mA h g⁻¹ retains after 60 cycles at 200 mA g⁻¹</td>
<td>80</td>
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<tr>
<td>CuO/GNS</td>
<td>Nanowire-on-sheet</td>
<td>A reversible capacity of 770 mA h g⁻¹ retains after 100 cycles at 100 mA g⁻¹</td>
<td>81</td>
</tr>
<tr>
<td>CuO/GNS</td>
<td>Sheet-on-sheet</td>
<td>An initial large charge capacity of ~1,092 mA h g⁻¹ at 70 mA g⁻¹, and the retained capacity of ~801 mA h g⁻¹ after 40 cycles</td>
<td>82</td>
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<tr>
<td>CuO/GNS</td>
<td>Fusiform-on-sheet</td>
<td>An initial large charge capacity of ~956 mA h g⁻¹ at 70 mA g⁻¹, and the retained capacity of ~666 mA h g⁻¹ after 40 cycles</td>
<td>82</td>
</tr>
<tr>
<td>CuO/GNS</td>
<td>Nanoleaf-on-sheet</td>
<td>A reversible capacity of 600 mA h g⁻¹ retains after 50 cycles at 100 mA g⁻¹</td>
<td>83</td>
</tr>
<tr>
<td>Fe₂O₃–N-doped graphene</td>
<td>Nanorod-on-sheet</td>
<td>A stable capacity of 508 mA h g⁻¹ at 2 A g⁻¹ after 200 cycles, and an impressive capacity of 249 mA h g⁻¹ at 20 A g⁻¹ after 2,000 cycles without capacity fading</td>
<td>84</td>
</tr>
<tr>
<td>Fe₂O₃/RGO</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large charge capacity of ~1,227 mA h g⁻¹ at 100 mA g⁻¹, and the retained capacity of ~1,027 mA h g⁻¹ after 50 cycles based on the mass of Fe₂O₃</td>
<td>85</td>
</tr>
<tr>
<td>Fe₂O₃/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large discharge capacity of ~1,320 mA h g⁻¹ at 0.1 C, and the retained capacity of ~650 mA h g⁻¹ after 50 cycles</td>
<td>86</td>
</tr>
<tr>
<td>Fe₂O₃/RGO</td>
<td>Nanoparticle-on-thin layer</td>
<td>A reversible capacity of 612 mA h g⁻¹ at 1 C, with a Coulombic efficiency of 98% after 50 cycles</td>
<td>87</td>
</tr>
<tr>
<td>Fe₂O₃/RGO</td>
<td>Nanoparticle-on-thin layer</td>
<td>A reversible capacity of 1,050 mA h g⁻¹ at 100 mA g⁻¹, and a reversible capacity of 500 mA h g⁻¹ at 1,600 mA g⁻¹</td>
<td>88</td>
</tr>
<tr>
<td>Fe₂O₃/GNS</td>
<td>Rice-on-sheet</td>
<td>An initial large discharge capacity of ~1,184 mA h g⁻¹ at 0.1 C, and the retained capacity of ~734 mA h g⁻¹ after 40 cycles</td>
<td>89</td>
</tr>
<tr>
<td>Fe₂O₃/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large discharge capacity of ~1,120 mA h g⁻¹ at 0.1 C, and the retained capacity of ~312 mA h g⁻¹ after 40 cycles</td>
<td>89</td>
</tr>
<tr>
<td>Mn₃O₄/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A high reversible specific capacity of more than 900 mA h g⁻¹ at 40 mA g⁻¹ with no capacity decay up to 50 cycles</td>
<td>90</td>
</tr>
<tr>
<td>MoO₃/GNS</td>
<td>Nanobelting-on-layer</td>
<td>A reversible capacity of 172 mA h g⁻¹ retained after 100 cycles at 100 mA g⁻¹</td>
<td>91</td>
</tr>
<tr>
<td>MoO₃/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large discharge capacity of ~1,296 mA h g⁻¹ at 0.1 C, and the retained capacity of ~1,330 mA h g⁻¹ after 100 cycles</td>
<td>92</td>
</tr>
<tr>
<td>ZnO/GNS</td>
<td>Nanocrystal-on-sheet</td>
<td>A reversible capacity of 460 mA h g⁻¹ retained after 50 cycles at 1 C</td>
<td>93</td>
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</table>

Abbreviations: GNS, graphene nanosheets; RGO, reduced graphene oxide.
Noerochim et al. and exhibits initial discharge capacity of 291 mAh g\(^{-1}\) at 100 mA g\(^{-1}\) and 172 mAh g\(^{-1}\) after 100 cycles. Moreover, 3D porous MoO\(_2\)/graphene microspheres were prepared in a microwave-assisted hydrothermal process. The obtained MoO\(_2\)/graphene composite exhibits excellent cycling stability of 1,300 mAh g\(^{-1}\) after 80 cycles at 0.1 C and good rate capability of 913 and 390 mAh g\(^{-1}\) at 2 and 5 C, respectively. The ZnO@graphene composite was synthesized from ZnO nanoparticles via a microwave-assisted deposition on GO in a microwave oven. It exhibits improved electrochemical performance with a high capacity of 850 mAh g\(^{-1}\) at 0.1 C. There is a small capacity decay of \(\sim 8\%\) during 50 cycles of discharge and charge.

**GNS-supported tin/germanium/silicon based anodes**

Tin, germanium and silicon are high-capacity elements for lithium storage, whose theoretical capacities values are 990,
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1,600, and 4,200 mAh g⁻¹, respectively. These theoretical values are calculated based on the lithium-ion reaction to form lithium alloys (LiₓSn, LiₓGe, and LiₓSi). It is worth noting that these elements are often used as oxides such as SnO, SnO₂, GeO₂, and SiO₂. The lithium ion can reduce these oxides to Sn, Ge, and Si at an early stage, followed by a similar lithium alloy and de-alloy storage mechanism. The oxygen element in these oxides is usually believed to be inactive for lithium ion storage. As summarized in Table 2, among the reported GNS-supported Sn nanoparticle anodes by microwave-assisted technology, Sn⁹⁴–⁹⁸ or SnO₂⁹⁸–¹⁰⁰ usually tends to form a nanoparticle/nanocrystal morphology in the presence of GNS. GNS-supported Ge nanoparticle¹⁰⁷ or thin film¹⁰⁸ and GNS-supported Si nanoparticle¹⁰⁹ or thin film¹¹⁰ was also synthesized by microwave irradiation or microwave plasma methods.

Figure 5A shows the schematic illustration of the preparation of graphene-supported Sn nanoparticles, which were synthesized via a single-mode microwave hydrothermal process in a microwave reactor, followed by hydrogen gas reduction. Figure 5B and C show the obtained GNS-supported Sn nanoparticles. Interestingly, the size of Sn nanoparticles in the Sn-GNS composite is changed from 60–120 nm (Sn-GNS-1) to 10–20 nm (Sn-GNS-2) when the ratio of Sn and GNS is reduced from 1:1 to 1:4. When used as the anode for LIBs, the Sn-GNS-1 and Sn-GNS-2 composites deliver reversible charge capacities of 1,206 and 1,407 mAh g⁻¹, respectively, with the corresponding Coulombic efficiencies of 67.9% and 65.9%. As shown in the cycling performance curves of Figure 5D, the Sn-GNS-1 and Sn-GNS-2 electrodes deliver specific capacities of 772 and 1,100 mAh g⁻¹, respectively, after 30 cycles at 0.1 C, which are both higher than that for bare GNS (582 mAh g⁻¹) after the same cycle number. The enhanced high-rate properties are also observed for Sn-GNS-2, which exhibits specific

Figure 3 The CuO-GNS composites and their electrochemical properties.
Notes: (A) Schematic illustration of the growth process of CuO-GNS sheet-on-sheet and fusiform-on-sheet structures, (B) TEM image of the CuO-GNS sheet-on-sheet composite, (C) TEM image of the CuO-GNS fusiform-on-sheet composite, (D) cycling performances at 0.1 C (70 mA g⁻¹) for various CuO-GNS composites, (E) rate capability performances of the CuO-GNS sheet-on-sheet composite. Reproduced from Lu LQ, Wang Y. Sheet-like and fusiform CuO nanostructures grown on graphene by rapid microwave heating for high Li-ion storage capacities. J Mater Chem. 2011;21:17916–17921, with permission from The Royal Society of Chemistry.²⁵
Abbreviations: GNS, graphene nanosheets; TEM, transmission electron microscopy.
discharge capacities of 1,247, 1,106, 946, and 876 mAh g\(^{-1}\) at current densities of 0.5, 1, 2, and 5 C. Moreover, a new strategy for the growth of self-assembled Sn@CNT on vertically aligned graphene (VAGN) was suggested by the microwave plasma irradiation method in MPCVD system.\(^{34}\) SnO\(_2\) is reduced to Sn on VAGN and subsequently encapsulated in the catalyzed carbon nanotubes (CNTs). In the Sn@CNT product, pear-like Sn core with a diameter of about 30 nm and a length of 40–50 nm is encapsulated inside a cylindrical CNT with a length less than 100 nm. The whole Sn@CNT structure is anchored on the surface of GNS. Such a composite exhibits a high reversible capacity of 1,026 mAh g\(^{-1}\) at 0.25 C, and a capacity of 140 mAh g\(^{-1}\) is retained in a short discharge time of 12 seconds. A Sn@graphene on the VAGN structure\(^{37}\) was also reported by the same authors as above, which exhibits a reversible capacity of 1,005 mAh g\(^{-1}\) at 0.25 C even after 120 cycles. Meanwhile, there are several other reports on GNS-supported SnO\(_2\) particles\(^{99–106}\) prepared by microwave-assisted method. Excellent electrochemical properties (a stable capacity of ~890 mAh g\(^{-1}\) without noticeable fading up to 80 cycles at 500 mA g\(^{-1}\)) were observed for the GNS-SnO\(_2\) nanocomposite, which was synthesized by a microwave-assisted hydrothermal process.\(^{101}\)

A unique sandwich-structured C/Ge/graphene composite\(^{107}\) was synthesized by a microwave-solvothermal process as shown in Figure 6A. A carbon coating layer is introduced on the surface of the graphene/germanium oxide composite precursor, followed by a reduction treatment. The composite consists of

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**Figure 4** The Fe\(_2\)O\(_3\)-GNS composites and their electrochemical properties. **Notes:** (A) Schematic illustration of the growth process of Fe\(_2\)O\(_3\)-GNS particle-on-sheet and rice-on-sheet composites. TEM images of (B) Fe\(_2\)O\(_3\)-GNS particle-on-sheet and (C) Fe\(_2\)O\(_3\)-GNS rice-on-sheet composite. Electrochemical performances of Fe\(_2\)O\(_3\)-GNS composites: (D) cycling performances at 0.1 C (100 mA g\(^{-1}\)) and (E) rate capability. Reprinted with permission from Zou YQ, Kan J, Wang Y. Fe\(_2\)O\(_3\)-graphene rice-on-sheet nanocomposite for high and fast lithium ion storage. J Phys Chem C. 2011;115:20747–20753. Copyright © 2011, American Chemical Society. **Abbreviations:** GNS, graphene nanosheets; TEM, transmission electron microscopy.
metalllic germanium nanoparticles (20–30 nm in size) between graphene sheets and carbon layers (Figure 6B and C). The C/Ge/graphene composite shows better cycling performances (a capacity of 993 mAh g\(^{-1}\) after 160 cycles, corresponding to 86.4% of the capacity at the second cycle) and rate capability (discharge capacity of 1,008 mAh g\(^{-1}\) after 30 cycles at 5 C) than the Ge/C and Ge/graphene composites.

3D graphene scaffold-supported Si thin film composite was prepared by Wang et al.\(^{110}\) As shown in Figure 7A, a 3D graphene scaffold is first synthesized using a MPCVD approach, and Si is then deposited on the graphene scaffold using radio frequency sputtering. The obtained Si grains (0.3–0.5 µm) are deposited on the surface of 3D graphene (Figure 7B). Figure 7C shows that these Si grains are composed of numerous Si nanoparticles (several nanometers in size). When used for LIBs, such 3D graphene scaffold-supported Si electrode exhibits an outstanding cycling stability. A capacity of 1,314 mAh g\(^{-1}\) can be observed after 500 cycles with capacity retention of 84% relative to the maximum capacity of 1,560 mAh g\(^{-1}\) in the 50th cycle. The composite also exhibits good high-rate cycling performances and a large capacity of 1,083 mAh g\(^{-1}\) is still retained after 1,200 cycles at a large current of 2.39 A g\(^{-1}\).

Based on the similar lithium alloy and de-alloy storage mechanism to Sn, Ge, and Si, Ag and Ag-based composites were also explored as the anodes for LIBs.\(^{111,112}\) It is found in the literature that only GNS-supported Ag nanoparticle\(^{111}\)/nanorod\(^{112}\) composites were synthesized by microwave-assisted methods and their electrochemical properties are presented in Table 2. The Ag nanorod on GNS\(^{112}\) is obtained

<table>
<thead>
<tr>
<th>Composites</th>
<th>Morphologies</th>
<th>Electrochemical performances</th>
<th>Reference</th>
</tr>
</thead>
<tbody>
<tr>
<td>Sn@CNTs-GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large discharge capacity of (-1,095) mAh g(^{-1}) at 200 mA g(^{-1}), and the retained capacity of (-1,013) mAh g(^{-1}) after 200 cycles</td>
<td>94</td>
</tr>
<tr>
<td>Sn/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>Charge capacities of 1,206 and 1,407 mAh g(^{-1}) with retained capacities of 772 and 1,100 mAh g(^{-1}) after 30 cycles at 0.1 C for Sn-GNS-1 and Sn-GNS-2, respectively</td>
<td>95</td>
</tr>
<tr>
<td>C-Sn/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A stable capacity of (-600) mAh g(^{-1}) at 100 mA g(^{-1}) with a higher capacity retention of 87% after 20 cycles</td>
<td>96</td>
</tr>
<tr>
<td>Sn/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large charge capacity of (-1,035) mAh g(^{-1}) at 150 mA g(^{-1}), and the retained capacity of (-1,005) mAh g(^{-1}) after 120 cycles</td>
<td>97</td>
</tr>
<tr>
<td>Sn/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>Discharge capacities of 1,300 and 730 mAh g(^{-1}) after 40 cycles for as deposited and annealed Sn-GNS, respectively</td>
<td>98</td>
</tr>
<tr>
<td>SnO(_2)/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>Capacities between 400 and 500 mAh g(^{-1}) after 80 cycles</td>
<td>99</td>
</tr>
<tr>
<td>SnO(_2)/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>Retained capacities of 590 and 504 mAh g(^{-1}) after 200 cycles for SnO(_2)-GNS1 and SnO(_2)-GNS2, respectively</td>
<td>100</td>
</tr>
<tr>
<td>SnO(_2)/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A stable capacity of about 890 mAh g(^{-1}) without noticeable fading up to 80 cycles at 500 mA g(^{-1})</td>
<td>101</td>
</tr>
<tr>
<td>SnO(_2)/GNS</td>
<td>Nanocrystal-on-sheet</td>
<td>An initial large charge capacity of (-1,329) mAh g(^{-1}) at 100 mA g(^{-1}), and the retained capacity of (-618) mAh g(^{-1}) after 20 cycles</td>
<td>102</td>
</tr>
<tr>
<td>SnO(_2)-RGO-CNT</td>
<td>Nanoparticle-on-sheet</td>
<td>A reversible capacity of the composite retains 502 mAh g(^{-1}) after 50 cycles at 100 mA g(^{-1}), and a capacity of 344 mAh g(^{-1}) at 1,000 mAh g(^{-1})</td>
<td>103</td>
</tr>
<tr>
<td>SnO(_2)/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large charge capacity of (-1,402) mAh g(^{-1}) at 100 mA g(^{-1}), and the retained capacity of (-1,359) mAh g(^{-1}) after 100 cycles</td>
<td>104</td>
</tr>
<tr>
<td>SnO(_2)/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A reversible capacity of 635 mAh g(^{-1}) retains after 100 cycles at 60 mA g(^{-1})</td>
<td>105</td>
</tr>
<tr>
<td>SnO(_2)/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A stable specific capacity of about 430 mAh g(^{-1}) retained after more than 140 cycles at 500 mA g(^{-1}) with a Coulombic efficiency close to 100%</td>
<td>106</td>
</tr>
<tr>
<td>C-Ge-GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A reversible capacity of 993 mAh g(^{-1}) retains after 160 cycles at 0.4 C</td>
<td>107</td>
</tr>
<tr>
<td>Ge/GNS</td>
<td>3D network</td>
<td>A reversible capacity of 1,140 mAh g(^{-1}) at 0.33 C over 100 cycles and 835 mAh g(^{-1}) at 8 C after 60 cycles</td>
<td>108</td>
</tr>
<tr>
<td>Si/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>Discharge capacity retention of about 800 mAh g(^{-1}) after 100 cycles at a current of 500 mA g(^{-1})</td>
<td>109</td>
</tr>
<tr>
<td>Si/GNS</td>
<td>Film-on-sheet</td>
<td>A capacity of 1,314 mAh g(^{-1}) after 500 cycles with capacity retention of 84% relative to the maximum capacity of 1,560 mAh g(^{-1}) in the 50th cycle</td>
<td>110</td>
</tr>
<tr>
<td>Ag/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>An initial large charge capacity of (-780) mAh g(^{-1}) at 0.1 C, and the retained capacity of (-714) mAh g(^{-1}) after 50 cycles</td>
<td>111</td>
</tr>
<tr>
<td>Ag/GNS</td>
<td>Nanorod-on-sheet</td>
<td>A high reversible capacity of 1,015 mAh g(^{-1}) at 0.1 C with a high capacity retention rate of 64.1% at 5 C</td>
<td>112</td>
</tr>
</tbody>
</table>

**Abbreviations:** CNTs, carbon nanotubes; GNS, graphene nanosheets; RGO, reduced graphene oxide.
by reducing Ag-GO hybrid. The Ag-GO precursor is prepared by two steps: Ag nanorods are first synthesized in a microwave solvothermal process in a household microwave oven and then mixed with GO prepared from Hummer’s method. The Ag-GNS nanorod-on-sheet composite shows a high reversible capacity of 1.015 mAh g\(^{-1}\) at 0.1 C with a high capacity retention rate of 64.1% at 5 C. By a similar microwave solvothermal process, GNS-Ag nanoparticle composite\(^{115}\) was synthesized from natural graphite and silver salt, and it exhibits an initial charge capacity of 780 mAh g\(^{-1}\), which decreases to 714 mAh g\(^{-1}\) after 50 cycles.

**GNS-supported metal sulfide anodes**

Metal sulfides are also promising high-capacity anode materials for LIBs. The active element for lithium ions can be either metal (e.g., Sn of SnS\(_2\)) or sulfur (e.g., S of CoS and NiS). Metal and sulfur are also suggested to be both active for In\(_2\)S\(_3\) materials, although most studies still believe the active element is only sulfur. The lithium ion storage mechanism of tin sulfide is similar to tin oxides, in which sulfide and oxygen are both inactive and lithium can reversibly react with tin to form Li\(_x\)Sn alloys (the maximum x value is 4.4). The storage mechanism of active sulfur element is based on the reversible formation and decomposition of lithium sulfide.

There are few reports on the GNS-supported metal sulfide composites with microwave-assisted syntheses: GNS-supported SnS\(_2\) nanosheets,\(^{115}\) SnS\(_2\) nanoparticles,\(^{69}\) In\(_2\)S\(_3\) particles and In\(_2\)S\(_3\) flowers\(^{114}\) composites were reported by Chen et al, Zou and Wang, and Gu and Wang, respectively. Their morphologies and lithium-storage properties are summarized in Table 3. A porous 3D SnS\(_2\)-reduced graphene oxide (RGO) sheet-on-sheet nanostructure was synthesized by a single-mode microwave solvothermal method at 180°C for 20 minutes in which SnS\(_2\) nanosheets are distributed uniformly on the RGO surface.\(^{115}\) If the amount of the starting GO is increased, the obtained SnS\(_2\) products are only nanoparticles.\(^{59}\) This is ascribed to the presence of a large amount of surface functionalities (mainly oxygen-containing groups) of GO, which can significantly affect the nucleation process of SnS\(_2\). As shown in Figure 8, large reversible capacities are observed from 1,077 to 896 mAh g\(^{-1}\) at 0.1 C and 934 to 657 mAh g\(^{-1}\) at 1 C in 40 cycles for the SnS\(_2\)-RGO sheet-on-sheet composite. Compared with bare graphene and pristine SnS\(_2\) nanoflowers, such a sheet-on-sheet composite achieves better electrochemical performances, which is attributed to the synergetic effect.
Microwave-assisted synthesis of graphene nanocomposites

**Figure 6** The C/Ge/graphene composite.

**Notes:** (A) Schematic illustration of the C/Ge/graphene composite, (B and C) TEM images of C/Ge/graphene composite at different magnifications. Reproduced from Li D, Seng KH, Shi DQ, Chen ZX, Liu HK, Guo ZP. A unique sandwich-structured C/Ge/graphene nanocomposite as an anode material for high power lithium ion batteries. *J Mater Chem A*. 2013;1:14115–14121 with permission of The Royal Society of Chemistry.

**Abbreviation:** TEM, transmission electron microscopy.

**Figure 7** The graphene supported Si composite.

**Notes:** (A) Schematic illumination of the growth process of the 3D graphene scaffold supported Si electrode (GSSSE). The as-synthesized GSSSE: (B) SEM image and (C) TEM image. Reproduced from Wang CD, Chui YS, Ma RG, et al. A three-dimensional graphene scaffold supported thin film silicon anode for lithium-ion batteries. *J Mater Chem A*. 2013;1:10092–10098 with permission of The Royal Society of Chemistry.

**Abbreviations:** SEM, scanning electron microscopy; TEM, transmission electron microscopy.
for highly reversible lithium-ion storage that resulted from the closely contacted sheet-on-sheet morphology. These electrochemical properties of the sheet-on-sheet composite are also superior to previous SnS$_2$-GNS particle-on-sheet composite in which an initial charge capacity of 858 mAh g$^{-1}$ is decreased to 652 mAh g$^{-1}$ after 40 cycles.

GNS-supported In$_2$S$_3$ nanoparticle and interconnected nanoflower composites were synthesized by the similar single-mode microwave hydrothermal method at 140°C under a pressure of 5.5 bars for 20 minutes. Black powders of In$_2$S$_3$-graphene particle-on-sheet composite are obtained instead of the tawny powders of In$_2$S$_3$-graphene flower-on-sheet when the dosage of graphene is increased. As indicated by Figure 9, In$_2$S$_3$ nanoparticles and nanoflowers are uniformly dispersed on GNS, forming sandwiched particle-on-sheet and unprecedented flower-on-sheet nanostructures. Compared with GNS and pristine In$_2$S$_3$, the GNS-supported In$_2$S$_3$ composites show extraordinary large reversible capacities and good cycling performances. Reversible initial capacities of 1,249, 913, 782, and 690 mAh g$^{-1}$ are observed at currents of 70, 700, 1,400, 3,500 mA g$^{-1}$, respectively, for the flower-on-sheet composite, while the particle-on-sheet composite shows slightly lower reversible capacities but more stable cycling performances at both small and high currents.

**Graphene-supported lithium titanium oxide based anodes**

Lithium-titanium-oxide-based materials are important high-rate anodes for lithium ion batteries, which are based on the lithium insertion mechanism of lithium storage. Although their theoretical capacities are even lower than graphite anodes, they exhibit very good cyclabilities because there is only a small volume change during the process of lithium insertion and extraction. A summary of morphologies and electrochemical performances of graphene-supported lithium-titanium-

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**Table 3** Summary on the morphologies and electrochemical performances of graphene-supported metal sulfides and lithium titanium-based anodes

<table>
<thead>
<tr>
<th>Composites</th>
<th>Morphologies</th>
<th>Electrochemical performances</th>
<th>Reference</th>
</tr>
</thead>
<tbody>
<tr>
<td>SnS$_2$/GNS</td>
<td>Sheet-on-sheet</td>
<td>Large reversible capacities of 1,077–896 mAh g$^{-1}$ at 0.1 C and 934–657 mAh g$^{-1}$ at 1 C in 40 cycles</td>
<td>113</td>
</tr>
<tr>
<td>In$_2$S$_3$/GNS</td>
<td>Flower-on-sheet</td>
<td>A reversible capacity of the composite retains 657 mAh g$^{-1}$ after 40 cycles at 70 mA g$^{-1}$</td>
<td>114</td>
</tr>
<tr>
<td>In$_2$S$_3$/GNS</td>
<td>Particle-on-sheet</td>
<td>A reversible capacity of the composite retains 614 mAh g$^{-1}$ after 40 cycles at 70 mA g$^{-1}$</td>
<td>114</td>
</tr>
<tr>
<td>Li$_5$Ti$_3$O$_7$/GNS</td>
<td>Microsphere-on-sheet</td>
<td>Capacities of 168, 161, 153, 147, 143, 132, 119, and 106 mAh g$^{-1}$ at 0.2, 0.4, 1, 2, 4, 10, 20, and 40 C, respectively</td>
<td>115</td>
</tr>
<tr>
<td>Li$_5$Ti$_3$O$_7$/RGO</td>
<td>Nanoparticle-on-sheet</td>
<td>A reversible capacity of 168 mAh g$^{-1}$ at 1 C with a high capacity retention rate of 59% at 50 C</td>
<td>116</td>
</tr>
<tr>
<td>Li$_5$Ti$_3$O$_7$/RGO</td>
<td>Nanoplatelet-on-sheet</td>
<td>A discharge capacity of 154, 128, and 101 mAh g$^{-1}$ at 1, 50, and 100 C, respectively</td>
<td>117</td>
</tr>
<tr>
<td>TiO$_2$/RGO</td>
<td>Microspheres-on-sheet</td>
<td>A large discharge capacity of 156 mAh g$^{-1}$ at 5 C and 84 mAh g$^{-1}$ retained at 60 C</td>
<td>118</td>
</tr>
</tbody>
</table>

**Abbreviations:** GNS, graphene nanosheets; RGO, reduced graphene oxide.

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The micro-wave hydrothermal method was reported to prepare Li$_5$Ti$_3$O$_7$ microspheres composed of nanoflakes wrapped in GNS. The obtained structure can avoid the restacking of GNS and offer rapid lithium diffusion; therefore, the composite exhibits highly desirable Li-ion storage properties in terms of a large capacity (168 mAh g$^{-1}$ at 0.2 C) approaching the theoretical value, stable cycling performance, and excellent rate capability. The Li$_5$Ti$_3$O$_7$/graphene composite was also prepared by the lithiation of the alkali titanate with the assistance of microwave. The composite exhibits a reversible capacity of 168 mAh g$^{-1}$ at a current rate of 1 C with a high capacity retention rate of 59% at a very large current rate of 50 C. Kim et al also reported Li$_5$Ti$_3$O$_7$ nanoplatelet/RGO hybrid, which was obtained from TiO$_2$/RGO nano-hybrid in LiOH aqueous solution via a microwave hydrothermal process. The composite can deliver a discharge capacity of 154, 128, and 101 mAh g$^{-1}$ at 1, 50, and 100 C, respectively. As reported by Yan et al., a facile microwave solvothermal process was developed to prepare an anatase TiO$_2$ anode material that maintains multiple properties including high surface area, high crystallinity, uniform mesoporous structure, perfect microspheres, and uniform particle size. Using this fine anatase TiO$_2$ product, a TiO$_2$/RGO hybrid material was prepared under UV-light irradiation. The incorporation of RGO improves the electrochemical kinetics of the TiO$_2$ microspheres (Figure 10), which results in superior electrochemical performance in terms of specific capacity, rate capability, and cycle stability. The lithium storage mechanism of the anatase TiO$_2$ is also a lithium insertion mechanism. A reversible lithium insertion and extraction reaction between TiO$_2$ and Li$_x$TiO$_2$ leads to a theoretical capacity of ~168 mAh g$^{-1}$. The composite also shows a large discharge capacity of 156 mAh g$^{-1}$ at a large...
current rate of 5 C. Even at 60 °C, a very high discharge capacity of 84 mAh g⁻¹ is still obtained.

**Other GNS composite anodes**

In the literature, there are also some reports about other types of GNS-based anode composites. Kang et al reported a CuCo₃O₄/RGO composite by a solvothermal reaction in a microwave reactor, followed by calcination treatment. Porous CuCo₃O₄ nanocubes are well wrapped by RGO sheets. Based on the observation of the Brunauer–Emmett–Teller (BET) results, the porous CuCo₃O₄ nanocube/RGO composite exhibits a higher specific surface area (34.4 m² g⁻¹)
than pristine CuCoO$_2$ nanocubes (10.9 m$^2$ g$^{-1}$). This would result in the increased contact areas between the electrode and the electrolyte solution when they are used as anodes for LIBs. Such a composite exhibits a high stable capacity of $\sim$570 mAh g$^{-1}$ at 1,000 mA g$^{-1}$ after 350 cycles. Impressive high-rate performance is also observed (a high capacity of $\sim$450 mAh g$^{-1}$ even at a high current density of 5,000 mA g$^{-1}$). The morphologies and electrochemical properties of other graphene-based anode materials$^{119-124}$ for LIBs synthesized via similar microwave hydrothermal/solvothermal process in microwave oven/reactor are summarized in Table 4. Multilayer GNS were prepared by a microwave hydrothermal technique and then mixed with single-walled carbon nanotube by vacuum filtering.$^{120}$ The obtained free-standing GNS-single-walled carbon nanotube film exhibits a large $d$-spacing of 0.41 nm and a reversible capacity of $\sim$300 mAh g$^{-1}$ during 50 cycles. Among these reports,$^{119-124}$ 3D GNS-CNT-Ni,$^{123}$ and GNS-CNT-Fe$^{24}$ composites show good lithium ion storage properties. They were both synthesized by similar microwave
hydrothermal method in a microwave oven/reactor, in which vertically aligned CNTs are grown directly on graphene sheets under the catalysis of Ni and Fe nanoparticles, respectively.

**GNS-decorated cathodes**

Graphene is an active anode for LIBs within a comparatively low voltage window, however there is almost no lithium storage capacity for graphene at a higher voltage for cathode. Therefore graphene has been investigated with more research concerns as anodes for LIBs. In comparison, only a small amount of graphene is used for cathode composites with the purpose to improve the electrical conductivity of the cathode. And the GNS-decorated cathodes prepared by microwave-assisted methods are summarized with their morphologies and electrochemical properties in Table 5. Among various graphene-based cathodes by microwave irradiation, graphene-LiFePO$_4$ composite has attracted more interests. 125–127 As shown in Figure 11, a GNS-supported LiFePO$_4$ nanorod composite was synthesized by a novel one-pot microwave solvothermal process within 15 minutes at a temperature below 300°C in an microwave synthesis system. 125 The obtained graphene/LiFePO$_4$ nanohybrid exhibits a discharge capacity of 164 mAh g$^{-1}$ (close to the theoretical value of 170 mAh g$^{-1}$) at 0.1 C and good cyclability up to 70 cycles. LiFePO$_4$/C/graphene composite was also reported by Shi et al. 126 After a rapid, one-pot, microwave-assisted hydrothermal method (15 minutes at 200°C), followed by sintering at 600°C for 2 hours under a H$_2$/Ar (5:95, v/v) atmosphere, the obtained LiFePO$_4$ particles have sizes around 150 nm. These particles are wrapped in crumpled micrometer-size graphene sheets. The LiFePO$_4$/C/graphene composite exhibits obviously improved electrochemical performance with highly stable reversible capacity of 88 mAh g$^{-1}$ at 1 C. Around 99% of the initial capacity can be retained after 40 cycles. Wang reported a similar LiFePO$_4$/C/graphene composite. 127 The composite was obtained via a direct solid-state heating reaction in a microwave oven, which delivers a large discharge capacity (157.8 mAh g$^{-1}$) at 0.1 C and more stable cycling performance than those of LiFePO$_4$/C. A LiFePO$_4$/RGO particle-on-sheet composite was synthesized via microwave hydrothermal method in a microwave oven. 128 The particle-on-sheet composite can deliver a reversible capacity of 138 mAh g$^{-1}$ at 0.1 C, and over 93.2% of its initial capacity can be retained after 100 cycles at 1 C. With the similar microwave hydrothermal method, the LiMn$_2$O$_4$/RGO nanoparticle-on-sheet composite

### Table 4 Summary on the morphologies and electrochemical performances of other graphene-based anode materials

<table>
<thead>
<tr>
<th>Composites</th>
<th>Morphologies</th>
<th>Electrochemical performances</th>
<th>Reference</th>
</tr>
</thead>
<tbody>
<tr>
<td>CuCo$_2$O$_4$/RGO</td>
<td>Nanocube-on-sheet</td>
<td>An initial large charge capacity of ~540 mAh g$^{-1}$ at 1 A g$^{-1}$, and the retained capacity of ~570 mAh g$^{-1}$ after 350 cycles</td>
<td>119</td>
</tr>
<tr>
<td>GNS-SWCNT</td>
<td>Free-standing film</td>
<td>An initial large charge capacity of 293 mAh g$^{-1}$ at 30 mA g$^{-1}$, and the retained capacity of 30 mAh g$^{-1}$ after 50 cycles</td>
<td>120</td>
</tr>
<tr>
<td>RGO-CNT</td>
<td>3D network</td>
<td>An initial charge capacity of 682 mAh g$^{-1}$ at 50 mA g$^{-1}$ and the retained capacity of 298 mAh g$^{-1}$ after 50 cycles</td>
<td>121</td>
</tr>
<tr>
<td>Zn$_2$GeO$_4$/N-doped graphene</td>
<td>Nanorod-on-sheet</td>
<td>An initial charge capacity of 873 mAh g$^{-1}$ at 100 mA g$^{-1}$ and the retained capacity of 1,044 mAh g$^{-1}$ after 100 cycles with an excellent rate capability (531 mAh g$^{-1}$ at 3.2 A g$^{-1}$)</td>
<td>122</td>
</tr>
<tr>
<td>3D GNS-CNT-Ni</td>
<td>3D network</td>
<td>An initial charge capacity of 1,089 mAh g$^{-1}$ at 100 mA g$^{-1}$ and the retained capacity of 648 mAh g$^{-1}$ after 50 cycles</td>
<td>123</td>
</tr>
<tr>
<td>3D GNS-CNT-Fe</td>
<td>3D network</td>
<td>A reversible capacity of ~1,024 mAh g$^{-1}$ after 40 cycles at 100 mA g$^{-1}$</td>
<td>124</td>
</tr>
</tbody>
</table>

**Abbreviations:** CNT, carbon nanotube; GNS, graphene nanosheets; RGO, reduced graphene oxide; SWCNT, single-walled carbon nanotube.

### Table 5 Summary on the morphologies and electrochemical performances of graphene-decorated cathodes

<table>
<thead>
<tr>
<th>Composites</th>
<th>Morphologies</th>
<th>Electrochemical performances</th>
<th>Reference</th>
</tr>
</thead>
<tbody>
<tr>
<td>LiFePO$_4$/GNS</td>
<td>Nanorod-on-sheet</td>
<td>A discharge capacity of ~164 mAh g$^{-1}$ at 0.1 C and good cyclability up to 70 cycles</td>
<td>125</td>
</tr>
<tr>
<td>LiFePO$_4$/C/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A high stable reversible capacity of 88 mAh g$^{-1}$ at 10 C and around 99% retained after 40 cycles</td>
<td>126</td>
</tr>
<tr>
<td>LiFePO$_4$/C/GNS</td>
<td>Nanoparticle-on-sheet</td>
<td>A high initial discharge capacity of 157.8 mAh g$^{-1}$ at 0.1 C and 94.7 mAh g$^{-1}$ at 5.0 C</td>
<td>127</td>
</tr>
<tr>
<td>LiFePO$_4$/C/RGO</td>
<td>Nanoparticle-on-sheet</td>
<td>A reversible capacity of 138 mAh g$^{-1}$ at 0.1 C, and retaining over 93.2% of its initial capacity after 100 cycles at 1 C</td>
<td>128</td>
</tr>
<tr>
<td>LiMn$_2$O$_4$/RGO</td>
<td>Nanoparticle-on-sheet</td>
<td>A high specific capacity of 137 mAh g$^{-1}$ at 1 C and a remarkably high discharge capacity of 117 mAh g$^{-1}$ and 101 mAh g$^{-1}$ at 50 and 100 C, respectively</td>
<td>129</td>
</tr>
<tr>
<td>FeF$_2$/RGO</td>
<td>Nanoparticle-on-sheet</td>
<td>A stable capacity of 150 mAh g$^{-1}$ retained after 50 cycles</td>
<td>130</td>
</tr>
</tbody>
</table>

**Abbreviations:** GNS, graphene nanosheet; RGO, reduced graphene oxide.
was obtained. The composite can deliver a high specific capacity of 137 mAh g⁻¹ at 1 C and a remarkably high discharge capacity of 117 mAh g⁻¹ and 101 mAh g⁻¹ at 50 and 100 C, respectively. Moreover, a FeF₃/RGO composite exhibits a nanoparticle-on-sheet morphology by a microwave solvothermal process. It delivers a stable capacity of 150 mAh g⁻¹ after 50 cycles when used as a cathode for LIBs.

Conclusion
Microwave synthesis has been demonstrated as a fast, uniform, energy-efficient, and scalable approach to prepare graphene-supported various electrodes. Representative examples such as graphene-supported transitional metal oxides, metal sulfide, tin/germanium/silicon/lithium titanium oxide based anodes, graphene-decorated lithium iron phosphate based cathodes, and some other graphene-based composite electrodes have been discussed. The fast microwave heating offers homogenous reaction environment and leads to good control of shape, size, size distribution, and agglomeration of the products. The surface functionalities on graphene can be controlled to different extents and the introduced second-phase component to graphene can be also tuned with 0D, 1D, 2Dmorphologies, and their stacked 3D network. These graphene-based composites usually exhibit strong synergetic effect when used for LIBs. They deliver larger capacity and better cyclability and high-rate performance compared to individual component of the composite. These improved electrochemical properties have been attributed to the preserved promising properties of graphene and the improved electrical conductivity and more stable mechanical structure of graphene-supported materials. These synthesized graphene composites with the assistance of microwave irradiation may find wide applications for other energy-storage applications such as supercapacitors and fuel cells. Furthermore, the microwave-assisted technology would be used more and more in the recent future to synthesize materials with controlled size and shape for the energy-storage application due to its simple, quick, inexpensive, uniform, and energy-efficient advantages.

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Disclosure
The authors report no conflicts of interest in this work.

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Figure 11 The LiFePO₄/GNS composite.

Notes: (A) TEM image, (B) HRTEM image, and (C) Elemental mapping images. Reproduced from Praneetha S, Murugan AV. A rapid, one-pot microwave-solvothermal synthesis of a hierarchical nanostructured graphene/LiFePO₄ hybrid as a high-performance cathode for lithium-ion batteries. RSC Adv. 2013;3:25403–25409 with permission of The Royal Society of Chemistry. Abbreviations: HRTEM, high resolution transmission electron microscopy; GNS, graphene nanosheets; TEM, transmission electron microscopy.


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