ORIGINAL RESEARCH

Influence of nanoparticles on color stability, microhardness, and flexural strength of acrylic resins specific for ocular prosthesis

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Department of Dental Materials and Prosthodontics, Araçatuba Dental School, São Paulo State University José Bonifácio, 1193, Vila Mendonça, Araçatuba, São Paulo, Brazil 16015-050 Fax +55 18 3636 3287 Email danielamicheline@foa.unesp.br Abstract: The aim of this study was to assess the effect of adding nanoparticles to N1 acrylic resin intended for artificial sclera, in terms of the color stability, microhardness, and flexural strength of the resin. Three hundred samples of N1 acrylic resin were used: 100 samples for color stability and microhardness tests (each test was performed on the opposite side of each sample), and 200 samples for flexural strength testing (100 samples before and after 1,008 hours of accelerated aging). Samples for each test were separated into ten groups (n=10), ie, without nanoparticles (control group) or with nanoparticles of zinc oxide, titanium dioxide (TiO₂), and barium sulfate at weight concentrations of 1%, 2%, and 2.5% (nanoparticle groups). Data were subjected to statistical analysis with nested analysis of variance and Tukey's test (P<0.05 significance level). Among the nanoparticle groups, the TiO, groups showed better color stability at all concentrations. Microhardness values increased after artificial aging, except for the control and zinc oxide groups. After aging, the 1%-2% TiO, groups had significantly higher microhardness values compared with the other nanoparticle groups. Before aging, there was a significant difference in flexural strength between the control and nanoparticle groups. After aging, the control and TiO₂ groups, regardless of concentration, showed the lowest flexural strength values. Incorporation of nanoparticles directly influenced the acrylic resin properties, with TiO₂ being the most influential nanoparticle in terms of the evaluated properties. Keywords: acrylic resins, eye, artificial, color, hardness, nanoparticles

Introduction

Ocular absence leads to difficulties in establishing social relationships, in addition to functional, emotional, personal, and esthetic disorders.¹ Ocular prostheses can restore and beautify a face with compromised expression, offering a safe, aesthetic, and satisfactory alternative to rehabilitate anophthalmic patients.^{2,3} Materials used for ocular rehabilitation demand excellent esthetics, durability, adaptation, and biocompatibility.^{4–7} For these reasons, white and colorless acrylic resins are the materials of choice for these prostheses.^{6–9}

One of the main reasons for refabrication of ocular prostheses is color degradation. Other problems affecting ocular prostheses include changes in the superficial microhardness and flexural strength over time, which directly influence material longevity. Currently, most of the research in this area focuses on optimizing prosthesis performance by improving the physical and mechanical properties of the prosthesis material. Several studies have found satisfactory results by incorporating nanoparticles (NPs) into polymeric materials (mainly silicone elastomers used for facial prostheses), in terms of protecting such materials from degradation.^{10–16} Research has

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© 2014 Andrexti et al. This work is published by Dove Medical Press Limited, and Licensed under Creative Commons Attribution — Non Commercial (unported, v3.0) permission from Dove Medical Press Limited, provided the work is properly attributed. Permissions beyond the scope of the License are administered by Dove Medical Press Limited. Information on how to request permission may be found at http://www.dovepress.com/permissions.php been directed towards the development of a new industrial process incorporating NPs into a polymeric matrix, with the goal of providing a new class of polymeric materials that offer the resistance of NPs and the flexibility of organic polymer matrices.^{17–19}

According to a survey of the literature, NPs that are widely used in industry include zinc oxide (ZnO), titanium dioxide (TiO_2) , and barium sulfate (BaSO₄). The addition of TiO₂²⁰ and ZnO to polymers can improve the mechanical and optical properties of polymers due to the small size, large specific area, and quantum effect of the NPs, as well as the strong interfacial interaction between the organic polymer and inorganic NPs.^{21,22} The NPs hardly scatter any visible light, being optically transparent, specially due to their nanometer scale and low content,²² in addition to exhibiting a high refractive index.²⁰ BaSO₄ is used in medical applications to examine the esophagus, stomach, and intestine by X-ray or computed tomography, as well as to confer higher radiopacity to rootend filling materials. In the cosmetic industry, BaSO₄ is used in sunscreen lotions to create a physical barrier against ultraviolet rays and to provide a better product appearance.^{15,16}

In addition to improving polymer properties, it has been suggested that incorporating NPs into an organic polymer could provide resistance to environmental stress-related cracking and aging.²³ However, to date, no study has been performed to examine the influence of NPs on the physical and mechanical properties of acrylic resins specific for artificial sclera. Thus, this study aimed to evaluate the influence of accelerated aging and different NP types (ZnO, TiO₂, and BaSO₄) and concentrations (1%, 2%, and 2.5%), in terms of the color stability, microhardness, and flexural strength of N1 acrylic resin for artificial sclera. The null hypothesis of the study was that addition of NPs would not influence the characteristics of acrylic resin according to the different types and concentrations of NPs used, and would be consistent over the periods studied.

Materials and methods Energy dispersive spectroscopy and scanning electron microscopy

To verify whether the acrylic resin contained any of the evaluated NPs in its composition, the chemical components of the resin were mapped through energy dispersive spectroscopy (EDS) in combination with scanning electron microscopy (SEM) analysis (EDS-SEM). Small volumes (1 μ m³) of the acrylic resin surface were characterized by EDS under 10,000× magnification, associated with SEM analysis of the images (JSM 610LA; JEOL Ltd, Tokyo, Japan).

Sample preparation

Three hundred samples of N1 acrylic resin (Artigos Odontológicos Clássico, São Paulo, Brazil) were manufactured. Of these 300 samples, 100 were used for the color stability and microhardness tests (each test was performed on the opposite side of the same sample) and 200 were used for the flexural strength tests (100 samples each for the tests before and after accelerated aging). Samples were divided into ten groups (n=10), according to the NP type and concentration that was combined with the N1 acrylic resin, ie, without NPs (control group) and 1%, 2%, or 2.5% ZnO, TiO₂, or BaSO₄ (for nine NP groups).²²

Samples in the color and microhardness tests were fabricated by a metallic matrix from silicone rubber disks (Zetalabor[®], Zhermack International, Rovigo, Italy) 3.0 cm in diameter and 0.3 cm in thickness.^{24,25} To create the molds, the disks were embedded into two-part flasks (Classico Dental Products, São Paulo, Brazil) with type IV dental stone (Durone, Dentsply, Petrópolis, Rio de Janeiro, Brazil). The flasks were maintained in a hydraulic press (Midas Dental Products Ltd, Sao Paulo, Brazil) for 2 minutes. After crystal-lization of the dental stone, the flasks were opened and the silicone disks removed.

Samples for the flexural strength test were fabricated from bars of colorless self-cured acrylic resin (Artigos Odontológicos Clássico) that were produced by a metal pattern. The bars were finished with metal abrasives of different grits (320, 600, 800, and 1,200; Buehler, Lake Bluff, IL, USA) in an automatic polisher (Ecomet 250 Pro/Automet 250, Buehler). Each final bar had a uniform texture and surface, with final dimensions of $64 \times 10 \times 3.3 \times 9$ mm.^{7,26} The bars were placed in microwave flasks. Type IV stone plaster and extra-hard laboratory silicone were used to facilitate sample deflasking. After plaster crystallization in the final inclusion, the flasks were opened, the bars were removed, and molds obtained.

The N1 acrylic resin was proportioned and manipulated according to the manufacturer's instructions. Briefly, Onda-Cryl polymer was used in an ideal volume ratio of three parts powder to one part liquid (21 mg to 7 mL). The mixture was mixed with a spatula for a few seconds in a glass pot, and the pot was closed. After the plastic phase was reached, resin was inserted into the molds in a microwave flask. NPs were incorporated at the moment of resin manipulation. The resin and NPs were weighed on a digital precision scale (BEL Equipamentos Analítico, São Paulo, Brazil), with NPs equivalent to 1%, 2%, or 2.5% of the acrylic resin weight. NPs at the three concentrations were hand-mixed with the acrylic resin powder, using a metal spatula in a glass pot.

A counter-flask was positioned and raised in a hydraulic bench press (Midas Dental Products Ltd) with a weight of 1.250 kgf. The counter-flask was maintained in place for 2 minutes. Bench polymerization was performed for 30 minutes. The resins were polymerized in a microwave oven (model BM535BBHNA, Maxi, Brastemp, São Paulo, Brazil), with the following technical specifications: power of 1,300 W, frequency of 60 Hz, and current of 20 A. The polymerization cycle was performed in a sequential treatment of 3 minutes at 30%, 4 minutes at 0%, and 3 minutes at 60% power, according to the manufacturer's instructions. After polymerization, the flasks were cooled naturally and opened. Samples were removed and finished with metallographic abrasives of 320, 600, 800, and 1,200 grit, simulating the laboratory conditions for making ocular prostheses. The dimensions of each sample were measured by a precision digital caliper (Meba, Zagreb, Yugoslavia).

Samples were stored in distilled water at 37°C for 24 hours (Cienlab, Campinas, São Paulo, Brazil) for hydration and residual monomer release. Thereafter, the samples were subjected to color stability, microhardness, and flexural strength tests, with each test being performed before and after 1,008 hours of accelerated aging.

Color stability test

Samples were submitted to initial chromatic analysis by an ultraviolet-visible reflection spectrophotometer (model UV-2450, Shimadzu Corporation, Kyoto, Japan). Color alterations were calculated with the L*a*b* system established by the Commission Internationale de l'Eclairage.²⁷ This system allows the mean value of color variation (ΔE) between two readings to be calculated by the formula: $\Delta E = [(\Delta L)^2 + (\Delta a)^2 + (\Delta b)^2]^{\frac{1}{2}}$.

Microhardness test

Knoop microhardness was determined according to American Society for Testing and Materials specification E384-11²⁸ using a microhardener (HMV-2T; Shimadzu Corporation) with a 25 g load for 10 seconds. For each sample, five measurements were performed. Each sample was placed 500 μ m from any other sample and 500 μ m from the sample boards.

Flexural strength test

This test was performed using the rupture modulus with a constant velocity of 5 mm/min until fracturing was detected. Samples were positioned in a universal testing machine (EMIC, São José dos Pinhais, São Paulo, Brazil) following

the three-point technique, with a distance of 50 mm between the sample and sides of the machine. A weight of 100 kg was applied, and the maximum fracture resistance was recorded in MPa.²⁹

Accelerated aging

After initial tests, accelerated aging was carried out in an aging chamber (Equilam, Diadema, Brazil), according to American Society for Testing and Materials specification G53.³⁰ Specimens were submitted to periods of alternating ultraviolet light and condensation of oxygen-saturated distilled water under conditions of heat and 100% humidity. Each aging cycle was performed over 12 hours. Specifically, the sample was irradiated with ultraviolet light for 8 hours at 60°C±3°C, and then condensation took place without light for 4 hours at 45°C±3°C. This process simulated the deterioration caused by rainwater, dew, and ultraviolet energy (UV-B) from direct and indirect sunlight. Samples were submitted to 1,008 hours of accelerated aging, and then new color, microhardness, and flexural strength tests were performed.

Statistical analysis

The effects of different types and concentrations of NPs on the acrylic resin properties were analyzed by nested analysis of variance. The concentration factor was nested in the NP factor because less than all concentration levels occur within each NP level. Significant differences were compared by Tukey's honestly significant difference test (α =0.05).

Results and discussion

The results indicated that the null hypothesis should be rejected. NPs directly influenced the acrylic resin properties, providing general higher color stability and microhardness values and lower flexural strength values, in comparison to the control group, after accelerated aging.

Elemental component analysis of the resin

Combined EDS-SEM analysis revealed that before NP incorporation, elements of the NPs were present in the composition of the acrylic resin polymer (Figures 1 and 2), although at extremely small amounts. Specifically, the EDS graphic (Figure 1) was predominated by peaks of carbon (70.37%) and oxygen (28.65%), which are present in polymethylmethacrylate molecules. Peaks of elements of the studied NPs represented 0.08% Ti, 0.01% Ba, and insignificant Zn. The minute amounts of these NPs in the acrylic resin justified the testing of higher concentrations of these NPs

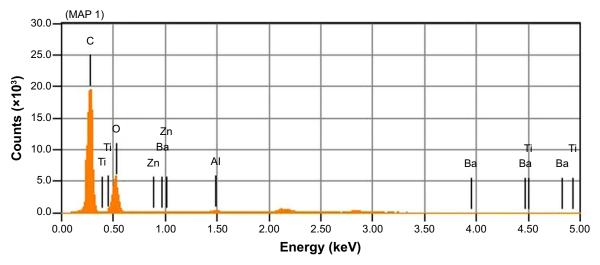


Figure I Energy dispersive spectroscopy analysis of acrylic resin polymer.

(1%–2.5%). The studied NP concentrations are consistent with those in other studies in the literature, especially Han et al who evaluated the effects of 0.5%-3% nano-oxides on the properties of a commercially available pigmented maxillofacial silicone elastomer.²²

Color stability

There were statistically significant differences in color stability for the factors of NP and concentration, with concentration nested in NP (Table 1). All samples showed considerable color changes (ΔE >0, Table 2). The control and BaSO₄ groups presented significantly higher values of color change compared with the other groups (Table 2).

Most polymers possess functional groups in their molecular chains, which are able to absorb ultraviolet light. When a polymeric molecule absorbs ultraviolet light, excessive energy leads to an unstable structural state. The excited molecule tends to disperse the excess energy, re-establishing

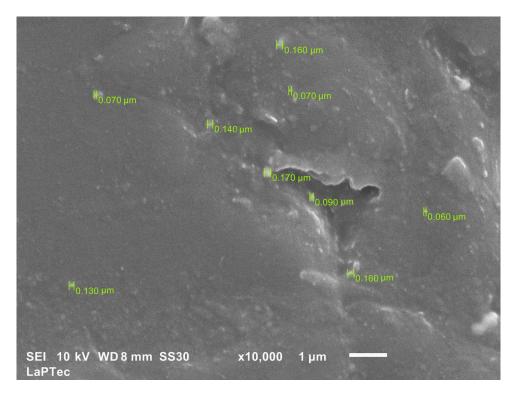


Figure 2 Scanning electron microscopy image of acrylic resin polymer. Note: Green highlights correspond to nanoparticles and their measurements. Abbreviations: SEI, secondary electron imaging; WD, working distance; SS, spot size.

Variation factor	df	SS sequential	SS adjusted	MS adjusted	F	P-value
Nanoparticle	3	405.4	405.4	135.1	330.7	<0.001*
Concentration (nanoparticle)	6	133.3	133.3	22.2	54.4	<0.001*
Error	90	36.8	36.8	0.4		
Total	99	575.4				

Table I Two-way nested analysis of variance for change in color (ΔE)

Note: *P<0.05 denotes statistically significant difference.

Abbreviations: df, degrees of freedom; SS, sum of squares; MS, mean of squares.

a stable structure. This dispersion event leads to rupture and photochemical degradation of the molecule, which contributes to material degradation, color or brightness changes, loss of opacity, cracks, and increased stiffness.^{31–33} Extrinsic factors (eg, absorption or adsorption of external substances)^{32,34} and intrinsic factors (eg, matrix alterations, which are very common with aging)³⁵ may also have contributed to the observed color changes.^{36,37}

In general, the samples incorporating NPs showed significantly greater color stability compared with the control group (Table 2). This finding is consistent with the results of other studies,^{10,11,38–40} which showed that addition of ZnO, TiO₂, or BaSO₄ to polymeric materials can increase the longevity of maxillofacial prostheses by promoting maintenance of color.^{10,12,38–40}

NPs are widely used, especially in sunscreen lotions, as inorganic absorbers of ultraviolet light.^{22,38,41,42} Inorganic NPs differ from some organic ultraviolet absorbers, in that they do not migrate in the polymeric matrix. Moreover, their photostability and thermal stability are not problematic, even over decades.⁴² NPs are smaller than the ultraviolet electromagnetic wave; thus, when ultraviolet light acts on NPs, part of the wave is dissipated while the rest is simultaneously absorbed. On the basis of these physical principles, protection may result from the absorption and dissipation of ultraviolet light by NPs, which act as a physical barrier by reflecting the light that reaches the prosthesis²² and preventing discoloration.^{40,41,43}

Different types and proportions of NPs were used in this study, with different levels of color stability, not all of which protected against color degradation of the resin (Table 2). These differences may be due to variations in the amount of organic impurities mixed with NPs after different handling processes by the manufacturers. Additionally, different NPs may interact differently with various polymer types.^{10,39} As the NP concentration increased, the ZnO and TiO₂ groups showed significant reductions in the color change values. Among the NP groups, the TiO₂ groups, especially the group with the 2.5% concentration, exhibited the least degree of color change (Table 2). Similarly, Han et al²² found that 2.5%

 ${\rm TiO}_2$ showed the best color stability results in their evaluation of nano-oxides incorporated into a facial silicone. Various studies have shown that ${\rm TiO}_2$ exhibits the best color protection results among studied NPs.^{10,12,40}

As a NP decreases in size, it offers increasing protection against ultraviolet light.^{22,44,45} Thus, we believe that particles of TiO_2 are smaller than those of ZnO or BaSO_4 , and that TiO_2 particles strongly link with acrylic resin particles to prevent resin degradation. In addition to NP size, the enhanced properties that arise from adding NPs to a polymer matrix may be attributed to the higher surface energy and chemical reactivity of the particles, which allow them to interact with the polymer matrix and form a three-dimensional network within the silicone polyethylene backbone.⁴⁶

According to the National Bureau of Standards, a color change is considered to be very low when ΔE is less than 1, clinically acceptable when ΔE is between 1 and 3, and clinically noticeable when ΔE exceeds 3.³⁹ In this study, the lowest mean color change value was 6.89, indicating that all of the evaluated groups had inappropriate degrees of color change.

Microhardness

Significant differences in microhardness were observed for all analyzed factors and interactions (Table 3). In general, the

Table 2 Average	values	of color	change	(ΔE)
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Concentration	Nanoparticle	Color change (∆E)
Not added	Control	13.74 (0.55) A
1%	ZnO	12.66 (0.45) B
	TiO ₂	10.22 (0.58) C
	BaSO₄	12.52 (0.45) B
2%	ZnO	10.10 (0.92) C
	TiO ₂	8.78 (0.61) D
	BaSO₄	13.38 (0.56) AB
2.5%	ZnO	9.14 (1.07) D
	TiO ₂	6.89 (0.38) E
	BaSO₄	14.02 (0.47) A

Notes: Data are shown as the mean (standard deviation). Different uppercase letters indicate statistically significant differences between the groups, by Tukey's test (P < 0.05).

Variation factor	df	SS sequential	SS adjusted	MS adjusted	F	P-value
Period	I	13.2	6.7	6.7	36.6	<0.001*
Nanoparticle	3	45.2	45.2	15.1	82.5	<0.001*
Period $ imes$ nanoparticle	3	37.9	38.0	12.6	69.3	<0.001*
Concentration (nanoparticle)	6	16.8	16.8	2.8	15.4	<0.001*
Period $ imes$ concentration (nanoparticle)	6	7.2	7.1	1.2	6.5	<0.001*
Error	180	32.8	32.8	0.2		
Total	199	153.0				

Table 3 Three-way nested analysis of variance for microhardness (Knoop)

Note: *P<0.05 denotes a statistically significant difference.

Abbreviations: df, degrees of freedom; SS, sum of squares; MS, mean of squares.

microhardness values increased with aging, except for the control and ZnO groups (Table 4). The increased microhardness after accelerated aging was probably a result of the complete resin polymerization that occurred when the samples were subjected to different types of energy (eg, high temperature and water condensation inside the aging chamber).^{47,48} After aging, the 1% and 2% TiO₂ groups exhibited significantly higher microhardness values compared with the other NP groups (Table 4). Similar to the color change results, this result can be explained by the presumably smaller size of TiO₂ compared with ZnO and BaSO₄, which would promote tight linking to the resin particles and prevent their degradation.^{22,44,45}

Specification 12 of the American Dental Association states that the surface microhardness of denture base acrylic resin must be at least 15 Knoop.^{24,49} As ocular prostheses are subjected to fewer impacts of lower intensity compared with denture bases, the microhardness values found in this study can be considered satisfactory and clinically acceptable.

Flexural strength

Statistically significant differences in flexural strength were observed in the interaction between the period and NP factors

 Table 4 Average values for microhardness (Knoop)

Concentration	Nanoparticle	Period			
		Initial	After aging		
Not added	Control	20.16 (0.35) ABa	20.04 (0.25) Ca		
1%	ZnO	20.12 (0.33) ABa	19.57 (0.57) CDa		
	TiO,	20.16 (0.44) ABa	20.93 (0.40) Bb		
	BaSO₄	19.26 (0.31) Ba	19.95 (0.23) Ca		
2%	ZnO	19.64 (0.29) ABCa	19.11 (0.78) Da		
	TiO,	20.31 (0.29) Aa	21.76 (0.34) Ab		
	BaSO₄	19.57 (0.32) Ba	20.81 (0.81) Bb		
2.5%	ZnO	19.50 (0.49) Ba	18.59 (0.28) Db		
	TiO,	19.88 (0.34) ABCa	20.84 (0.40) Bb		
	BaSO₄	19.22 (0.36) BCa	21.36 (0.38) ABb		

Notes: Data are shown as the mean (standard deviation). Different uppercase letters indicate statistically significant differences between the groups, while different lowercase letters indicate statistically significant differences between the periods, by Tukey's test (P<0.05).

(Table 5). Before aging, the NP groups at all concentrations showed significantly lower flexural strength values compared with the control group (Table 6). Most likely, the incorporation of NPs in the resin matrix hindered cross-linking of the polymer chains and had a degrading effect.³³ Furthermore, more so than any other property examined in this study, flexural strength is directly influenced by NP type.⁵⁰ Other possible reasons for the decreased flexural strength with incorporation of NPs include the increased porosity with NP addition and agglomerations around the incorporated NPs, which favors fracture.⁵¹

There was a significant difference in flexural strength between the periods before and after aging for the control and TiO₂ groups. Addition of TiO₂ resulted in lower flexural strength values after aging, most likely due to the nature of this NP (Table 6). Despite the very small size of TiO₂ NPs, they are difficult to disperse in organic solvents and tend to agglomerate easily.⁵⁰ The agglomerations can reach micrometer scales that are detrimental to flexural strength, because large porous areas may appear around them, increasing the likelihood of fracture.⁵²

The increased risk of fracture is one reason why using a higher NP concentration was not feasible in the current study. Although NPs can reinforce the polymer matrix, the filler content must be maintained at an appropriate level because of the higher surface energy and chemical reactivity of the NPs.⁴⁶ If not appropriately maintained, the NPs may agglomerate. When the acrylic resin is under external forces, the agglomerated particles act as stress-concentrating centers in the resin matrix, thereby decreasing its mechanical strength.¹⁷

Han et al observed that nano-oxide particles incorporated in silicone elastomer at a concentration of 2.0% maintained their original size and improved the mechanical properties of silicone. However, when the concentration was increased to 3.0%, the SEM images demonstrated agglomeration of all the nano-oxides. Nevertheless, incorporation of nano-oxides improved a wide range of mechanical properties when the NP

Variation factor	df	SS sequential	SS adjusted	MS adjusted	F	P-value
Period	I	13,014.3	15,065.5	15,065.5	96.9	<0.001*
Nanoparticle	3	3,329.7	3,329.7	1,109.9	7.1	<0.001*
Period $ imes$ nanoparticle	3	8,141.4	8,141.4	2,713.8	17.5	<0.001*
Concentration (nanoparticle)	6	503.5	503.5	83.9	0.5	0.78
Period $ imes$ concentration (nanoparticle)	6	1,052.8	1,052.8	175.5	1.1	0.35
Error	180	27,999.4	27,999.4	155.6		
Total	199	54,041.1				

 Table 5 Three-way nested analysis of variance for flexural strength (MPa)

Note: *P<0.05 denotes a statistically significant difference.

Abbreviations: df, degrees of freedom; SS, sum of squares; MS, mean of squares.

concentration ranged between 2.0% and 2.5% by weight.¹⁷ Overall, our findings and the observations from previous studies suggest that these NPs require some surface modification to reduce agglomeration and improve their dispersion in resin matrix.⁵⁰

We also believe that the increased microhardness due to incorporation of NPs may have reduced the flexural strength. As the microhardness increases, a material becomes more friable, and therefore more susceptible to fracture. According to International Organization for Standardization 1567, the flexural strength of microwave-polymerized acrylic resin should be at least 65 MPa.⁵³ Although the flexural strength values decreased with addition of NPs, in this study, all of the values for flexural strength were satisfactory and clinically acceptable.

Summary

The samples were subjected to 1,008 hours of accelerated aging, equivalent to one year of clinical use.^{16,54} Ultraviolet light, humidity, and heat caused polymer degradation by altering the color stability, microhardness, and flexural strength of N1 acrylic resin. Accelerated aging may have caused degradation of the organic resin matrix, involving hydrolysis and decomposition of the polymeric chain.^{32,40,55} The importance of maintaining and improving the properties of acrylic resin for ocular prosthesis use are well known. NPs are increasingly being used to improve the performance of

Table 6 Average values of flexural strength (MPa) for nanoparticle

 type and period, independent of concentration

Nanoparticle type	Period				
	Before aging	After aging			
Control	112.23 (9.04) Aa	76.45 (17.11) ABb			
ZnO	84.85 (15.34) Ca	80.65 (12.31) Aa			
TiO,	97.08 (13.35) Ba	66.91 (11.21) Bb			
BaSO ₄	83.31 (10.62) Ca	75.82 (9.96) Aba			

Notes: Data are shown as the mean (standard deviation). Different uppercase letters indicate statistically significant differences between the groups, while different lowercase letters indicate statistically significant differences between the periods, by Tukey's test (P<0.05).

polymeric materials. Therefore, more studies are required to elucidate the effects of NPs, and the mechanisms underlying these effects, on the physical and mechanical properties of prosthesis materials.

Conclusion

Considering the limitations of an in vitro study, it can be concluded that:

- Accelerated aging, through ultraviolet light, humidity, and heat, was a causative factor of polymer degradation, altering the properties of N1 acrylic resin
- Incorporation of NPs, especially 2.5% TiO₂, helped in maintaining color stability and microhardness after aging
- Flexural strength was relatively impaired by addition of NPs, although the results were within the clinically acceptable standard.

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Disclosure

The authors report no conflicts of interest in this work.

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